Liquid-State Processes

- In Situ composites
- Reinforcements are formed during the solidification of the molten metal within the matrix, e.g., controlled unidirectional solidification of eutectic alloys.



Fig. 6.9 Transverse sections of in situ composites obtained from a eutectic at different solidification rates indicated in *left-hand top* comers (cm/h). The nickel alloy matrix has been etched away to reveal the TaC fibers. [From Walter (1982), used with permission]

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Some of the more important processes:

✓ Diffusion bonding

- Used to join similar or dissimilar metals
- Predetermined stacking of:

✓ Matrix alloy foil and fiber arrays
✓ Composite wire
✓ Monolayer laminates

• Simultaneous application of pressure and high temperature

→ Inter-diffusion of atoms from clean metal surfaces in contact at elevated temperature



Fig. 6.8 (a) Schematic of diffusion bonding process.

(b) Microstructure of Ti/SiC_{fiber} composite made by diffusion. (900°C, 105 MPa, 3 h, fiber diameter =142 μ m) **Composite Materials, 2014, BN, IUT, Iran**



✓ The main advantages:

•The ability to process a wide variety of matrix metals

•Control of fiber orientation and volume fraction

× The main disadvantages:

- •Processing times of several hours
- •High processing temperatures and pressures
- •Quite expensive
- •Only objects of limited size can be produced

✓ Deformation processing of metal/metal composites

•Less conventional: In-situ composite

•Extrusion, drawing, rolling or ... of a ductile two-phase material.

•The two phases co-deform \rightarrow the minor phase elongates and becomes fibrous within the matrix

•The starting material is usually a billet prepared by casting or powder metallurgy methods.

•More conventional:

- **Roll bonding** to produce sheet laminated MMCs
- ARB (accumulative roll bonding)
- CAR (continual annealing and roll-bonding)



Schematic illustration of ARB and CAR processes Fig. 2

Composite Materials, 2014, BN, IUT, Iran R. Jamaati et al., J Mat Eng and Perform 21(7), 2012, 1249-1253.

Nb–Ti composite superconductors ✓ Extremely fine Nb–Ti superconducting filaments embedded in a copper matrix



Fig. 9.6 (b) One strand containing 1,060 filaments (Dia. = $50 \mu m$)



Fig. 9.7 SEM of Cu/Nb-Ti superconductor



Fig. 9.5 Fabrication route for Nb-Ti/Cu composite superconductors

✓ Annealed Nb–Ti rods are inserted into hexagonal-shaped high purity copper tubes.

 \checkmark These rods are loaded into a copper tube, evacuated, sealed, and extruded.

 \checkmark The extruded rod is cold drawn and annealed repeatedly to the appropriate final size and properties.

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fabricating Cu/Nb₃Sn superconductor

Fig. 9.9 (a) An Nb₃Sn/Cu composite superconductor. Each little dot in the picture is a 4- μ m dia. Nb₃Sn filament. (b) SEM of a Nb/bronze composite before the heat treatment to form Nb₃Sn.

• Deposition techniques for MMC fabrication

1- Coating individual fibers in a tow with the matrix material

2- Diffusion bonding to form a consolidated composite plate or structural shape

•Several deposition techniques are available:

- ✓Immersion plating
- ✓Electroplating
- ✓ Spray deposition
- ✓ Chemical vapor deposition (CVD)
- ✓ Physical vapor deposition (PVD)

More information available in the text book.

Mechanical Properties

• Young's Modulus

- ✓ Unidirectionally reinforced continuous fiber reinforced MMCs:
 - A linear increase in the longitudinal Young's modulus as a function of the fiber volume fraction
 - The modulus increase in a direction transverse to the fibers is very low.



Fig. 6.19 Properties of Al_2O_3/Al -Li composites as a function of fiber volume fraction (V_f): (a) axial and transverse Young's modulus vs. fiber volume fraction, (b) axial and transverse ultimate tensile strength vs. fiber volume fraction. [From Champion et al. (1978), used with permission]

Mechanical Properties

- Particle reinforced MMCs :
 - Increased modulus
 - The stiffness enhancement in particulate composites is reasonably isotropic



Fig. 6.20 Increase in Young's modulus of an MMC as a function of reinforcement volume fraction for continuous fiber, whisker, or particle reinforcement

Properties

> Strengthening mechanisms in metal matrix composites

> 1- Direct strengthening:

- Considers only the contribution of the reinforcement and the matrix (load transfer from the matrix to high modulus reinforcements)
- Is critically dependant on the reinforcement-matrix interface
- Does not take into account any strength contribution from microstructural changes in the metal matrix
- For fiber reinforced MMCs:
 - Rule of mixtures: $\sigma_{\rm c} = \sigma_{\rm f} V_{\rm f} + \sigma_{\rm m} V_{\rm m}$
 - $\sigma =$ the stress
 - V = the volume fraction
 - c, f, and m denote the composite, fiber, and matrix, respectively

• For particle reinforced MMCs:

$$\Delta \sigma_l = v_p \sigma_m \left[\frac{(l+t)A}{4l} \right]$$

 $\Delta \sigma_l$ = load-bearing contribution of reinforcement

- v_p = volume fraction of particles in the matrix
- σ_m = yield strength of the matrix
- *l* = size of the particulate parallel to the load direction

$$t$$
 = thickness of the particulate, and

$$A = l/t = particles aspect ratio$$

•For equiaxed (spherical) particles:

$$\Delta \sigma_l = 0.5 v_p \sigma_m$$

A. Sanaty-Zadeh, P.K. Rohatgi, Materials Science and Engineering A 531 (2012) 112–118.

> 2- Indirect strengthening:

- The reinforcement-induced changes in matrix microstructure and properties including:
 - Orowan strengthening
 - Grain and substructure strengthening
 - Quench hardening
 - Work hardening
 - Solid solution strengthening
- The indirect strengthening appears to be more important in particle reinforced composites.

> I- Orowan strengthening

- Hard and non-shearable particles may pin and block the dislocations.
- Orowan effect=*Gb/l*
 - G = shear modulus of the matrix
 - *b* = Burgers vector of the matrix
 - *l*= particle spacing
- The degree of strengthening is believed to be insignificant for microsized reinforcements.
- Nano MMCs seem to benefit more from this mechanism.

> II- Grain and substructure strengthening

• Hall–Petch relationship:

$$\sigma_y = \sigma_0 + \frac{k_y}{\sqrt{d}}$$

- σ_y = yield strength
- σ_0 = a materials constant (resistance of the lattice to dislocation motion)
- $k_v =$ the strengthening constant
- d = grain or sub-grain size in the matrix.

• Grain boundary strengthening can be high in spray cast and powder metallurgy processed composites.

A356-2%CNT



Fig.5 Typical micrographs of cast samples: (a) A-0-0; (b) A-0-2; (c) A-15-0; (d) A-15-2; (e) A-30-0; (f) A-30-2

Abasipour, B., Niroumand, B., Monir-Vaghefi, M., Transactions of Nonferrous Metals Society of China, 20, (2010) 1561-1566.



Fig. 3. Influence of reinforcement particle vol% on the matrix grain size refinement of Mg MMNCs, (a) influence of Y_2O_3 on pure Mg MMNCs, and (b) influences of SiO₂ and SiC on alloy Mg MMNCs.

J.B. Ferguson, F. Sheykh-Jaberi, C.S. Kim, P.K.Rohatgi, K. Cho, Materials Science & Engineering A, 558 (2012) 193–204.



Fig. 7. Influence of Al₂O₃, SiO₂, and SiC concentrations and inverse square root of grain size on the (a) yield strength (σ_y) and (b) strain to failure (ε_f) of Mg alloy MMNCs. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)

J.B. Ferguson, F. Sheykh-Jaberi, C.S. Kim, P.K.Rohatgi, K. Cho, Materials Science & Engineering A, 558 (2012) 193–204. Composite Materials, 2014, BN, IUT, Iran

> III- Quench hardening

- Reinforcements normally have smaller coefficient of thermal expansions (CTEs) than the matrix
- When subjected to a temperature change, thermal stresses are generated in both of the components
- A metal matrix undergoes plastic deformation in response to the thermal stresses generated and thus alleviates them
- A high density of dislocations may be generated around the reinforcements

Alumina fiber/Mg alloy matrix

Twins in Mg matrix → plastic deformation in Mg due to thermal stresses

RZ= reaction zone



•A dislocation etch-pitting technique was used to delineate dislocations in single crystal copper matrix (1975).



Fig. 6.16 A primary plane section of a metal matrix composite is shown as having a hard zone (high dislocation density) around each fiber and a soft zone (low dislocation density) away from the fiber



Fig. 6.26 Dislocation distribution in the aluminum matrix of a SiC_w/Al composite: (a) inhomogeneous dislocation distribution before testing, (b) uniform dislocation distribution after fatigue testing. [From Williams and Fine (1985a), used with permission]

- Thermal strain in the matrix: $e_{\rm m} = \Delta \alpha \Delta T$
- The dislocation density resulting from CTE mismatch:

$$\rho_{\rm CTE} = \left(AeV_{\rm p}\right)/b\left(1-V_{\rm p}\right)d$$

- A = a geometric constant
- e = the thermal misfit strain
- b = the Burgers vector
- V_p = the particle volume fraction
- *d* = the particle diameter
- The strength contribution is given by

$$\sigma_{\rm q} = \alpha G b (\rho_{\rm CTE})^{1/2}$$

 α = a constant

G = the shear modulus of the matrix

- Accelerated aging processes due to heterogeneous nucleation at dislocations
- This contribution of quench hardening to strength can be significant.

> IV- Work hardening

- Due to the modulus mismatch (strain misfit) between the elastic reinforcement and the plastic matrix
- Reinforcements affect the matrix work hardening rate.
- ✓ Effect of modulus mismatch on the strength of a composite

$$\sigma_d = \sqrt{3}\alpha Gb\sqrt{\rho^{\rm EM}}$$

- $-\alpha = a \operatorname{constant} (\sim 0.5)$
- ρ^{EM} = the dislocation density caused by modulus mismatch
- G = the shear modulus of the matrix
- b = the Burger's vector

$$\rho^{\rm EM} = \frac{6\nu_p}{\pi d_p^3}\varepsilon$$

*v*_p= volume fraction of particles
d_p = the particle diameter
ε= the uniform deformation

A. Sanaty-Zadeh, P.K. Rohatgi, Materials Science and Engineering A 531 (2012) 112-118.

> V- Solid solution strengthening

• The reinforcements may affect the microsegregation as a result of solute segregation at the interfaces or chemical reaction with the matrix.

 \rightarrow The content and distribution of solute in the matrix is changed

- In MMCs reinforced with continuous fibers, direct strengthening is a major factor.
- For discontinuously reinforced metals, quench hardening and work hardening are likely to be the most active mechanisms.
- For nano-sized reinforcements, Orowan strengthening may become a key mechanism.
- Normal matrix strengthening due to solution and precipitation hardening and grain refinement will give additional strength to the composite.
- + The strength of MMCs is most strongly dependent on the volume fraction of reinforcement.

The overall strength of the composite:

•There are different models. Two examples:

1- Simply add up all the strengthening contributions:
•Neglects the effect of different mechanisms on each other
•Assumes that each mechanism behaves independently

$$\sigma = \sigma_0 + \Delta \sigma_1 + \Delta \sigma_2 + \Delta \sigma_3 + \Delta \sigma_4 + \cdots$$

2- Clyne method (for micro composites):

$$\sigma_y = \sigma_0 + \Delta \sigma$$

 $\Delta \sigma = \sqrt{(\Delta \sigma_l)^2 + (\Delta \sigma_{\text{Orawan}})^2 + (\Delta \sigma_{\text{Hall-Petch}})^2 + (\Delta \sigma_{\text{EM}})^2 + (\Delta \sigma_{\text{CTE}})^2 + (\Delta \sigma_{\text{WH}})^2}$

 σ_0 = Yield strength of the unreinforced matrix

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•An expression for the yield strength of a particulate composite:

$$\sigma_{\rm yc} = \sigma_{\rm ym} [1 + (L+t)/4L] V_{\rm p} + \sigma_{\rm ym} (1 - V_{\rm p})$$

 $\sigma_{\rm ym}$ = the yield stress of the unreinforced matrix,

- Vp = the particle volume fraction,
- L = the length of the particle perpendicular to the applied load and
- t = the length of the particle parallel to the loading direction



Fig. 7. Strengthening mechanism contributions as a function of volume fraction for the particle sizes 10 nm and 1 µm.

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- For Mg matrix reinforced with nano-Al₂O₃ and Y_2O_3 particles
 - \checkmark There is a 75 nm particle size threshold
 - ✓ Larger particle sizes do not significantly influence the strength of the nanocomposite.

• Theoretical calculation in Prof. Rohatgi's group:

✓ An ideal Al-15vol% Al_2O_3 (10 nm particles) ✓ → 1GPa yield strength!