

CHAPTER 4

Interfaces

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Interface

- An interface between any two phases: a surface across which a discontinuity in one or more of material parameters or properties occurs.
 - For example in lattice parameter, density, elastic modulus, thermal expansion coefficient, strength, fracture toughness, etc.
- The behavior of a composite material is a result of the combined behavior of the:
 - Reinforcement
 - Matrix
 - Reinforcement/matrix interface

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Interfacial area

✓**Example:** Calculate the total reinforcement/matrix interface area (per cm^3) of a 50% fiber reinforced composite ($d=10 \mu\text{m}$).

- For a cylindrical fiber in a matrix:
 - Surface area per unit volume of a fiber (ignoring the ends)

$$\frac{S}{V} = \frac{2}{R} = \frac{4}{d} \quad (d = \text{diameter})$$

- Fine fibers (diameter \approx a few μm) can lead to very large interfacial areas
- For a laminated composite made by laminating sheets of two materials of thickness t :
 - The interfacial area $\propto 1/t$
- It easily can go as high as $3,000 \text{ cm}^2/\text{cm}^3$ in composites

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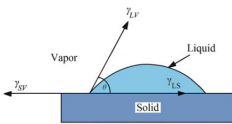
- The reinforcements should not be weakened by flaws because of an adverse interfacial reaction
- The applied load should be effectively transferred from the matrix to the reinforcements via the interface
 - The nature of the interface region under a given set of conditions
 - Wettability of the reinforcement by the matrix
 - The type of bonding between the two components
 - How the characteristics of the interface are affected by temperature, diffusion, residual stresses, and so on.

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Wettability

Definition: Ability of a liquid to spread on a solid surface

- The sessile drop experiments: A liquid drop will spread and wet a surface only if this results in a net reduction of free energy of the system.
- Equilibrium of three forces (three specific surface tensions or energies):
 - γ_{sv} of the solid/vapor interface
 - γ_{ls} of the liquid/solid interface
 - γ_{lv} of the liquid/vapor interface



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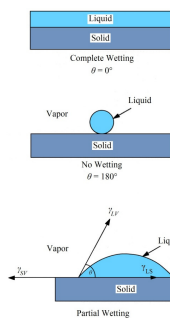
Contact angle, θ

- Young's equation: Equilibrium of forces in horizontal direction gives

$$\gamma_{sl} + \gamma_{lv} \cos \theta = \gamma_{vs}$$

$$\text{or, } \cos \theta = (\gamma_{vs} - \gamma_{sl}) / \gamma_{lv}$$

- $\theta = 0^\circ$, perfect wetting
- $\theta = 180^\circ$, no wetting
- For $0^\circ < \theta < 180^\circ$, partial wetting



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Contact angle

- **Wettability \neq Bonding**
- A low contact angle, i.e. good wettability, is a necessary but not sufficient condition for strong bonding.
- Contact angle (θ) is also a function of:
 - Interfacial reactions and contamination
 - Time and temperature of contact
 - Substrate roughness and geometry
 - Humidity
 - Environment
 - ...

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Effect of Surface Roughness

- The interface between the reinforcement and matrix is never perfectly planar
- Most reinforcements show some degree of roughness

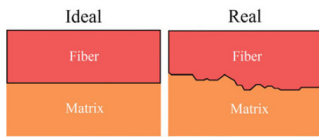


Fig. 4.2 (a) An ideal planar interface between reinforcement and matrix. (b) A more likely jagged interface between fiber and matrix

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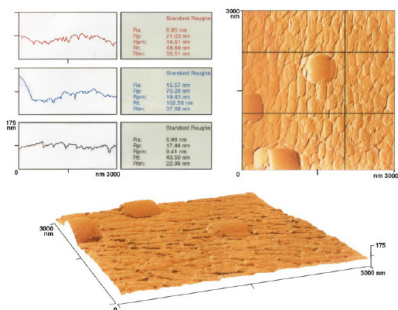


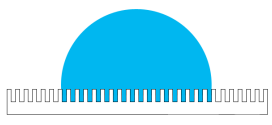
Fig. 4.3 Characterization of surface roughness of a polycrystalline alumina fiber (Nextel 610) by atomic force microscopy. The three profiles on the left-hand side correspond to the two horizontal and one vertical lines on the right-hand side figure. The bottom figure shows a three-dimensional perspective view of surface

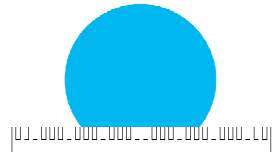
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- A good wetting is required for the liquid to penetrate the crevices.
- The effect of surface roughness on wettability can be described by *Wenzel's equation*:

$$\cos \theta_0 = r \frac{\gamma_{SV} - \gamma_{LS}}{\gamma_{LV}}$$
- $r = A_{real} / A_{proj}$
 - A_{real} = the real area of the interface
 - A_{proj} = the projected area of the interface.
$$\cos \theta_{Wenzel} = r \cos \theta_{Young}$$
- If $0 \leq \theta_Y < 90 \rightarrow$ wettability is enhanced by roughness.
- If $90 < \theta_Y \leq 180 \rightarrow$ wettability is reduced by roughness.
- If wetting is poor ($\theta > 90$), surface roughness can reduce bonded area and lead to void formation.

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- Wenzel model: 

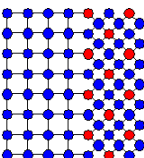
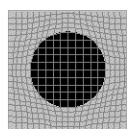
- Cassi-Baxter model: 

[Next assignment!](#)

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Crystallographic Nature of Interface

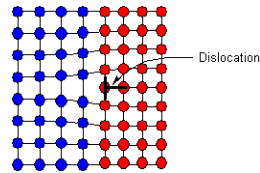
- ✓ **Coherent interface:**
 - One-to-one correspondence between lattice planes on the two sides of the interface
- Involves an elastic deformation of the crystals
- Has a lower energy than an incoherent one
 - Example: The interface between GP zones and the Al matrix in Al-Cu system

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✓ **Semicoherent interface:**

- Only a partial atomic registry
- Small lattice mismatch between the two phases accommodated by the introduction of dislocations at the interface.

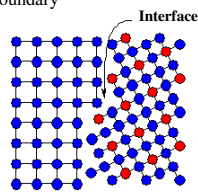


- Examples: Interfaces between a precipitate and a matrix , interfaces in some eutectic composites such as NiAl–Cr system

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✓ **Incoherent interface:**

- Severe atomic disorder, no matching of lattice planes across the boundary, no continuity of lattice planes across the interface
- No coherency strains, but high boundary energy because of severe atomic disorder at the grain boundary



- *Most of the interfaces that one encounters in fiber, whisker, or particle reinforced composites are incoherent.*

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Interactions at the Interface

- Components of composites are rarely in thermodynamic equilibrium.
 - A driving force for some **interfacial reaction(s)** leading to a state of thermodynamic equilibrium
- An initially *planar interface* can become an *interfacial zone* with multiple interfaces resulting from the formation of different intermetallic compounds, interdiffusion, and so on.
- Thermodynamic information can help **predict** the final equilibrium state of the composite:
 - Phase diagrams, reaction kinetics, diffusivities of one constituent in another, ... can provide information about the rate at which the system would tend to attain the equilibrium state.

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- In the absence of thermodynamic and kinetic data, **experimental studies** have to be done to determine the compatibility of the components.
- Characteristics of the interfacial zone (different phases in the interfacial zone):
 - Compositional parameter
 - Geometry and dimensions
 - Microstructure and morphology
 - Mechanical, physical, chemical, and thermal characteristics
 - ...

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Interface Reactions of Aluminum with Carbon Fibers

Carbon fiber / Al Si6 - interface (TEM analysis):

SEM of a Carbon fiber after thermal exposition of Al-MMC (650°C, 10 min) and matrix removal with sodium hydroxide

Possible reaction products and precipitates:

- Al_4C_3
- Mg_2Si
- Al_3MgO_4
- others, depending on alloying elements
- free Si

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Interface Reactions with Ceramic Reinforcements

Chemical interface reaction products in Al Mg4 / SiC_p (Squeeze Casting):

- Al_4C_3
- Mg_2Si

→ Cu-alloying (AlCuMgAg matrix, T6) yields best mechanical results ($\sigma_s = 700$ MPa, $K_{IC} = 9.5$ MPa m^{1/2}) due to load dissipation by ductile matrix.

Reaction products after thermal exposition for ceramic oxide (Al_2O_3) fibers:

loss of mech. properties

$MgAl_2O_4$ crystals on Nextel 610 fiber surface (extracted from EN AW-6061, heat treated at 540°C / 16 h; courtesy of 3M)

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- **Fabrication process** can alter the interface characteristics.
 - For example, the difference in the CTEs of the two components results in thermal stresses at the interface
- In MMCs
- The softer component (generally the matrix) will deform plastically.
 - High dislocation density observed in the matrix near the interface
 - Examples: Cu/W_{fiber} and Al/SiC_{whisker} cast composites and many other MMCs

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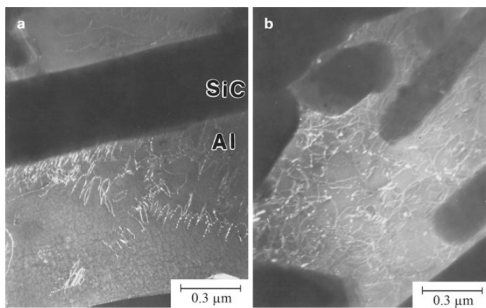


Fig. 6.26 Dislocation distribution in the aluminum matrix of a SiC_w/Al composite: (a) inhomogeneous dislocation distribution before testing, (b) uniform dislocation distribution after fatigue testing. [From Williams and Fine (1985a), used with permission]

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- In PMCs and CMCs
- The matrix is unlikely to deform plastically in response to the thermal stresses → microcracking
- In powder processing techniques
 - The nature of the powder surface (e.g. an oxide film always present on the surface of powder particles) influences the interfacial interactions and chemical nature of the powder.
 - **Topographic characteristics** of the components
 - Affect the degree of atomic contact between the components
 - Geometrical irregularities at the interface (roughness, voids, ...) → Stress concentrations

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Types of bonding at the interface

✓ Important types of interfacial bonding:

- Mechanical bonding
- Physical bonding
- Chemical bonding

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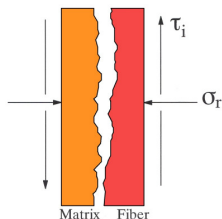
Types of bonding at the interface

• **Mechanical Bonding**

- Mechanical keying or interlocking can lead to bonding.
- Matrix in a composite radially shrinks more than the reinforcement on cooling from a high temperature.
→gripping of the reinforcement by the matrix even in the absence of any chemical bonding.
- The matrix penetrating the crevices on the reinforcement surface, by liquid or viscous flow or high temperature diffusion, can also lead to some mechanical bonding.

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Mechanical bonding



Mechanical (or Frictional) Bond

σ_r : the radial gripping stress
 τ_i : the interfacial shear stress
 μ : the coefficient of friction (0.1-0.6)

$$\tau_i = \mu \sigma_r$$

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Mechanical bonding

- In general, mechanical bonding is a low energy bond compared to chemical bonding
 - i.e., the strength of a mechanical bond is lower than that of a chemical bond.
- Pure mechanical bonding alone is not enough in most cases.
 - It could add, in the presence of reaction bonding, to the overall bonding
- Mechanical bonding is efficient in load transfer when the applied force is parallel to the interface
 - The matrix must fill the hills and valleys on the surface of the reinforcement

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Mechanical keying: Importance of wettability

- Mechanical keying of the matrix depends on the roughness of the reinforcement.

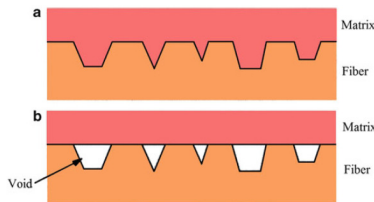


Fig. 4.5 (a) Good mechanical bond. (b) Lack of wettability can make a liquid polymer or metal unable to penetrate the asperities on the fiber surface, leading to interfacial voids

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Mechanical keying

- Surface roughness can contribute to bonding only if the liquid matrix wets the reinforcement surface.
- If the matrix (liquid polymer or metal) is unable to penetrate the asperities on the fiber surface, then the matrix will leave interfacial voids on solidification !
- An example of excellent wetting is between WC and cobalt liquid ($\theta = 0$)

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Examples of mechanical bonding

- Carbon fiber/epoxy
 - Nitric acid oxidation of carbon fibers
 - Increases specific surface area + good wetting
 - Improved interlaminar shear strength (ILSS) of the composite
- Al₂O₃ / Al composites:
 - There is only mechanical bond between Al₂O₃ and Al.
 - Rough interface: More efficient load transfer from the aluminum matrix to the alumina
- CMCs:
 - Mechanical bonding is preferred over chemical bonding!
 - Roughness-induced gripping at the interface is quite important.

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Physical bonding

- Any bonding involving
 - Van der Waals forces,
 - Dipolar interactions, or
 - Hydrogen bonding
- Bond energy is very low,
E ~8–16 kJ/mol.

Chemical bonding

- Covalent,
- Ionic, or
- Metallic bonding
- Bond energy is high,
E ~ 40 - 400 kJ/mol

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Chemical bonding

- Chemical bonding involves:
 - Atomic or molecular transport by diffusional processes,
 - Solid solution and compound formation at the interface,
 - Formation of a interfacial reaction zone having a certain thickness,
 - High energy covalent, ionic, and metallic bonds
- Two main types of chemical bonding:
 - Dissolution bonding
 - Reaction bonding

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Chemical Bonding: *Dissolution Bonding*

- Atomic species dissolve into one another at the interface.
- Short-range interaction (at an electronic scale)
- An excellent wettability is of great importance.
- In the absence of an intimate contact, the characteristic short-range interaction cannot occur.
- Surfaces should be treated to remove any impurities, contamination or entrapped air or gas bubbles at the interface

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Chemical bonding: *Reaction bonding*

- Solid solutions and compound formation in the interface zone.
- Atomic, ionic or molecular transport by diffusional processes from one or both of the components to the reaction site (interface)
- Wettability is again of great importance

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Interfacial reactions in MMC

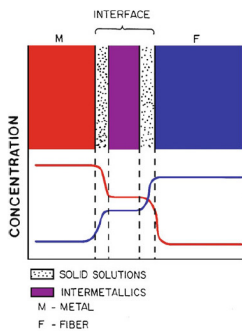


Fig. 4.6 Interface zone in a metal matrix composite showing solid solution and intermetallic compound formation

The reaction products and the reaction rates can vary depending on the matrix composition, reaction time, and temperature.

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Solid solution and compound formation at the interface

- Diffusion theory: For a diffusion-controlled growth

$$x^2 \sim Dt$$

$$D = A \exp(-Q/kT)$$

x = reaction zone thickness,
 D = diffusion coefficient,
 t = time,
 A = pre-exponential constant,
 Q = activation energy,
 k = Boltzmann's constant
 T = temperature in K

$$t \uparrow \rightarrow X \uparrow$$

$$T \uparrow \rightarrow X \uparrow$$

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Solid solution and compound formation at the interface

- Processing of (cast) MMCs: Both **T** and **t** are generally large
 → Significant chemical reaction which may adversely affect the behavior of the composite
- Examples: **Al-SiC**
 $4 Al + 3 SiC \leftrightarrow Al_4C_3 + 3 Si$
- → Formation of aluminum carbide (brittle)
- Addition of Si to the matrix (change in melting point, solidification mode, properties, ...)
- Control of time, temperature, and initial composition
- High Si content (>10%) alloys are invulnerable

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Solid solution and compound formation at the interface

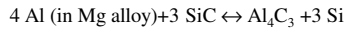
Al-Mg/Al₂O₃

- High levels of Mg in the matrix: MgO formation at the interface
 $3 Mg + Al_2O_3 \leftrightarrow 3 MgO + 2 Al$
- Low levels of Mg in the matrix: Spinel formation at the interface
 $3 Mg + 4 Al_2O_3 \leftrightarrow 3 MgAl_2O_4 + 2 Al$
- *Controlled amount of reaction* at the interface → Strong interfacial bonding
- Too thick an interaction zone → Adverse effects on properties

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Solid solution and compound formation at the interface

Mg-Al/SiC



- Formation of aluminum carbide and injection of Si to the matrix → Formation of Mg_2Si in the matrix

Ti-6Al-4V/SiC

TiC and Ti_xSi_y formation on the interface

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Optimum Interfacial Bond Strength

- Maximizing the bond strength is not always the goal!
- In CMCs, too strong a bond would cause embrittlement.
- If the interface is as strong or stronger than the reinforcement, it will have the lowest strain-to-failure of the three components (i.e. reinforcement, matrix, and interface)
- The composite will fail when any cracking occurs at a weak spot along the brittle interface → very low toughness

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Optimum Interfacial Bond Strength

- Optimum interfacial bond strength: Enhanced toughness, but without a severe penalty on the strength
- Such a composite will have multiple failure sites spread over the interfacial area, which will result in a diffused or global spread of damage, rather than a very local damage.

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Tests for Measuring Interfacial Strength

Numerous tests have been devised including:

- Flexural Tests
 - Three and Four-Points Bending
 - Short-Beam Shear Test (Interlaminar Shear Stress Test)
 - Iosipescu Shear Test
- Single Fiber Pullout Tests
- Curved Neck Specimen Test
- Instrumented Indentation Tests
- Fragmentation Test
- Laser Spallation Technique

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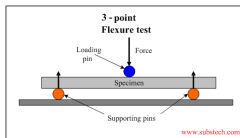
Tests for Measuring Interfacial Strength

Flexural Tests: Very easy to do

- **Three-Point Bending Test:**
 - Bent until the interface fails
 - Maximum tensile stress occurs at the outermost surface.

$$\sigma = \frac{3PS}{2bh^2} \quad (\text{Eq. 1})$$

P = load, S = span,
b = specimen width, h = specimen height



- Fibers parallel to the specimen length
→ measures longitudinal strength of fiber/matrix interface
- Fibers perpendicular to the specimen length
→ measures transverse strength of fiber/matrix interface.

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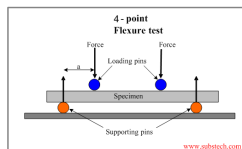
- Maximum shear stress, τ_{max} , occurs at the midplane and is given by

$$\tau_{max} = \frac{3P_{max}}{4bh} \quad (\text{Eq. 2})$$

- **Four-Point Bending Test:**

$$\sigma = \frac{3PS}{4bh^2}$$

S = the outer span



- No transverse shear stresses occur on the cross sections of the beam between the two inner loading points

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**Short beam shear test
(Interlaminar Shear Stress test - ILSS)**

- A special longitudinal three-point bend test with fibers parallel to the length of a very *small bend samples* ($S = 5h$), ASTM (D2344), can also be used for *laminar composites*

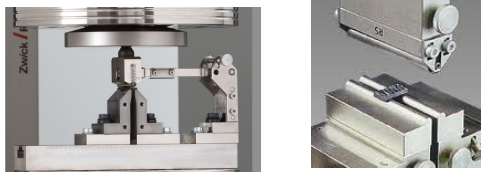
- Dividing max. shear stress (Eq. 2) by max. tensile stress (Eq. 1), we get

$$\frac{\tau}{\sigma} = \frac{h}{2S}$$

- Important message:
 - Make load span, S , very small \rightarrow Maximize shear stress, τ
 - \rightarrow Short specimens are more likely to fail in shear
 - Specimen fails under shear with a crack running along the mid-plane.

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**Short beam shear test
(Interlaminar Shear Stress test - ILSS)**

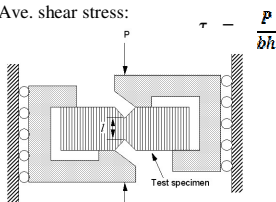
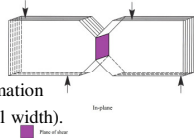


- If the fibers fail in tension before shear-induced failure occurs or if shear and tensile failure occur simultaneously \Rightarrow **Invalid** test!
- Examine the fracture surface after the test to ensure that the crack is along the interface and not through the matrix!
- For advantages/disadvantages refer to the text book.

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Iosipescu shear test

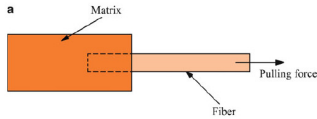
- Double edge-notched sample
- A special type of four-point bend test
- Off-set rollers \rightarrow Accentuates shear deformation
- Notch angle=90, Notch depth=22 % of full width).
- Uniform pure shear in the plane
- Ave. shear stress:



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Single fiber pullout test

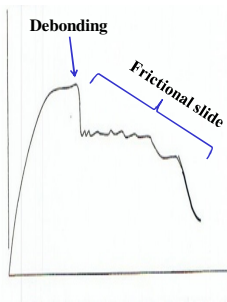
- Can provide useful information about the interface strength in model composite systems.
- Not very useful in case of commercially available composites.



- A portion of fiber, length l , is embedded in a matrix and a pulling tensile force is applied.
- Fabrication of the single fiber pullout test sample is often the most difficult Part.

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Single fiber pullout test



- The interfacial shear strength is a function of the coefficient of **friction** and any **normal compressive stress** at the interface.
- The main source of **radial compressive stress** is the **shrinkage of the matrix** during cooling from the processing temperature.
- Effect of different **Poisson's contractions** of fiber and matrix can result in a **radial tensile stress** at the interface (see Chap. 10).

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Single fiber pullout test

- The stress required to pull the fiber out of the matrix as a function of the embedded fiber length is measured.
- Must avoid any fiber misalignment and introduction of bending moments.

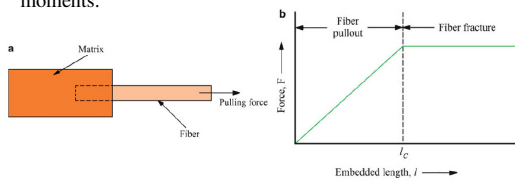


Fig. 4.8 (a) An experimental setup for a single fiber pullout test. A portion of fiber, length l , is embedded in a matrix and a pulling tensile force is applied as shown. (b) The stress required to pull the fiber out of the matrix as a function of the embedded fiber length

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Single fiber pullout test

- The stress required to pull the fiber out without breaking it increases linearly with the embedded fiber length, up to a critical length, ℓ_c .
- For embedded fiber lengths $\geq \ell_c$, the fiber will fracture under the action of the tensile stress, σ , acting on the fiber.

- The tensile stress, σ , acting on the fiber results in a shear stress, τ , at the fiber/matrix interface.
- Force balance along the fiber length gives

$$\sigma \pi r^2 = \tau 2\pi r \ell$$

- For $\ell < \ell_c$, the fiber is pulled out and the interfacial shear strength is given by

$$\tau = \sigma r / 2\ell$$

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Instrumented Indentation Tests

- Pointed, conical or rounded indenters can be used to displace a fiber aligned perpendicular to the composite surface.

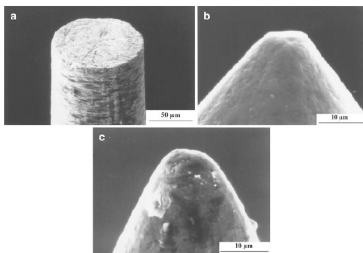


Fig. 4.9 Three different types of indenters: (a) cylindrical with a flat end, (b) conical with a flat end, and (c) pointed (courtesy of J. Janczak-Rusch and L. Rohr)

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Instrumented Indentation Tests

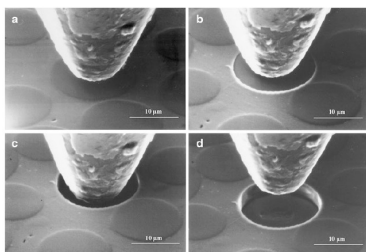


Fig. 4.10 An example of fiber pushin is shown in a series of four photographs. (a) Indenter approaching the fiber (b) touching the fiber (c) fiber pushin (d) lifting of indenter (courtesy of J. Janczak-Rusch and L. Rohr)

- By measuring the applied force and displacement, interfacial stress can be obtained.

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Instrumented Indentation Tests

- The specimen thickness must be large compared to the fiber diameter, e.g. 1-3 mm.

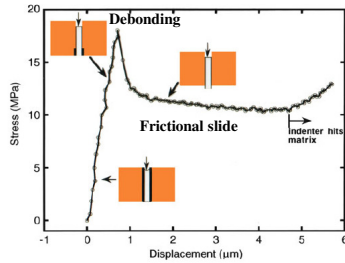


Fig. 4.11 Stress vs. displacement obtained in fiber pushout test (Chawla et al. 2001)

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Instrumented Indentation Tests

- Three-regions** in a valid pushout tests curve:
 - 1st region:** The indenter is in contact with the fiber and the fiber sliding is less than the specimen thickness.
 - 2nd region:** A horizontal region in which fiber sliding length is greater than or equal to the sample thickness.
 - 3rd region:** The indenter comes in contact with the matrix.
- In the first region, the fiber is elastically compressed by the indenter load over the debonded length, which is assumed to be dependent on the interfacial friction.
- The axial load on the indenter is assumed to be balanced by the frictional stress at the interface, and the effect of radial expansion during indentation is neglected.

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Instrumented Indentation Tests

- In the horizontal region, the interfacial shear stress is given by:

$$\tau_i = \frac{P}{2\pi r t}$$

P: the applied load
r: the fiber radius
t: the specimen thickness

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Other Tests

- Curved Neck Specimen Test
- Fragmentation Test
- Laser Spallation Technique

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